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ARTICLE OPEN

Asymmetric photocurrent extraction in semitransparent laminated flexible organic solar cells

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Scalable production methods and low-cost materials with low embodied energy are key to success for organic solar cells. PEDOT (PSS) electrodes meet these criteria and allow for low-cost and all solution-processed solar cells. However, such devices are prone to shunting. In this work we introduce a roll-to-roll lamination method to construct semitransparent solar cells with a PEDOT(PSS) anode and a polyethyleneimine (PEI) modified PEDOT(PSS) cathode. We use the polymer:PCBM active layer coated on the electrodes as the lamination adhesive. Our lamination method efficiently eliminates any shunting. Extended exposure to ambient degrades the laminated devices, which manifests in a significantly reduced photocurrent extraction when the device is illuminated through the anode, despite the fact that the PEDOT(PSS) electrodes are optically equivalent. We show that degradation-induced electron traps lead to increased trap-assisted recombination at the anode side of the device. By limiting the exposure time to ambient during production, degradation is significantly reduced. We show that lamination using the active layer as the adhesive can result in device performance equal to that of conventional sequential coating.

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INTRODUCTION

There is an urgent need to replace the present fossil-based energy production with sustainable energy sources. Thin-film solar cells and organic photovoltaics (OPV) in particular are technologies with a large potential to contribute to the energy transition due to very short energy payback times¹ and scalable production methods.² The best devices approach 12% power conversion efficiency (PCE)³ and due to improved OPV performance under low light-intensity conditions the yearly-average energy output is improved, reducing the competitive gap between OPV and silicon photovoltaics. Nevertheless, the OPV technology is not intended to compete with the high PCE of silicon-based solar cell installations but is instead targeted for entry markets that require the unique features of OPV, such as curved form factors enabled by flexible substrates, color tunability, semitransparency, low weight and the uniquely low energy payback times.

To realize these advantages materials and production methods must be scalable with high throughput, such as roll-to-roll printing under ambient atmosphere. Vacuum-based processing, commonly used to deposit reflective (such as aluminum/chromium or silver) or transparent metal electrodes (indium tin oxide (ITO)), should ideally be avoided due to their high energy input.⁴ Indium in ITO is in addition too scarce to be a scalable alternative. Solution-processed electrodes have a significantly lower embodied energy and allow for the high speed printing needed for scalable production. To fully utilize the benefits of solution processing all layers in the device should be coated or printed.

The use of semitransparent Poly(3,4-ethylenedioxythiophene)-poly(styrenesulfonate), PEDOT(PSS) electrodes for both the anode

and the cathode allows for the rapid printing of OPV modules at a small energy budget and a potentially short energy payback time. Highly conductive PEDOT(PSS) dispersions have been proven to be a sustainable substitute for ITO^{5–7} as the transparent electrode. The high work function of PEDOT(PSS) makes it a suitable anode without further modification. The work function of PEDOT(PSS) can be modified, converting it to a cathode electrode, e.g., by a layer of for example ZnO⁸ or a thin layer of polyethyleneimine (PEI)⁹ on top of PEDOT(PSS). Solution processable, flexible and semitransparent PEDOT(PSS) electrodes are thus a promising candidate for the upscaling and industrialization of OPV.

However, solution-cast solar cells with both electrodes being PEDOT(PSS) typically suffer from a low manufacturing yield, mainly due to a high probability of short-circuiting (or shunting),¹⁰ an issue we have experienced in the past.^{11,12} The shunting problem arises following the wet deposition of the top PEDOT(PSS) electrode and is avoided when the top electrode is evaporated. The actual mechanism leading to this shunting behavior is presently under discussion.^{8,10} By applying a high reverse bias pulse the shunts can be eliminated and the device in most cases starts functioning. Shunting may be caused by the contact between the top and the bottom electrodes,¹⁰ or due to a non-selective PEDOT(PSS) anode. The bias pulse is suggested to electrochemically de-dope the PEDOT(PSS) anode to switch it to an electron blocking layer.⁸ The exact mechanism for the shunting behavior remains unclear. Regardless of the origin, shunting adds a possibly rate-limiting step to the manufacturing process¹³ and limits the geometric fill factor (FF) for metal-free modules, as all

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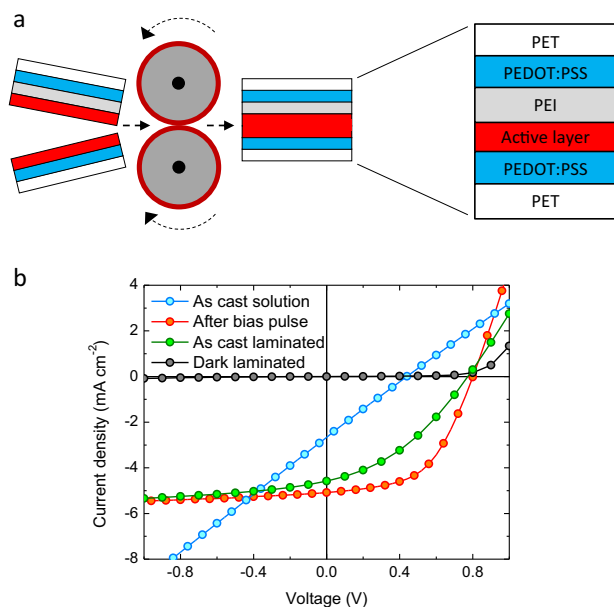


Fig. 1 Laminated solar cells **a** Schematic of roll-to-roll lamination using the active layer as the adhesive. **b** IV curves for laminated (green), laminated dark current (gray) and sequentially solution-cast TQ1:PC₇₁BM solar cells as cast (blue) and following a reverse bias pulse (red)

subcells must be accessible for the application of the electric pulse.

Lamination, in this respect, can be a useful method to avoid short-circuits, as it avoids the wet deposition of the top electrode. Lamination of electronic devices based on soft organic materials was demonstrated with the development of poly(3-alkylthiophenes) around 1987.¹⁴ Lamination of PEDOT(PSS) electrodes on top of the active layer using adhesive additives has also been used to construct OPV devices.^{15–19} Active layer lamination has previously been demonstrated for polymer/polymer blends²⁰ and also polymer/PCBM bilayers,²¹ but with poor PCE below 0.1%.

In this work we show that the polymer:PCBM bulk heterojunction (BHJ), coated on flexible PEDOT(PSS) electrodes, works as an excellent lamination adhesive to generate well performing solar cells, removing the need for adhesive additives. We laminate an active layer (AL1) coated on a semitransparent anode to another active layer (AL2) coated on a semitransparent cathode using a scalable roll lamination method. By lamination we obtain all solution-processed solar cells without any shunts. The laminated devices are unaffected by electric pulses, which shows that non-shunted solar cells with PEDOT(PSS) as both electrodes can be produced if wet deposition of the top electrode is avoided. We have laminated a number of different polymer:PCBM material systems to show the generality of the lamination method, mainly focusing on blends of the polymer poly(2,3-bis-(3-octyloxyphenyl) quinoxaline-5,8-diyl-*alt*-thiophene-2,5-diyl) (TQ1)²² with PC₇₁BM, enabling PCE of ~6 % in devices with an evaporated reflective electrode. TQ1 has also been our first choice when upscaling synthesis and deposition methods for roll-to-roll printing methods.

Unfortunately, ambient processing causes degradation of laminated devices, severely limiting their performance. This degradation manifests itself as a surprising asymmetry in the extracted photocurrent, depending on whether the device is illuminated through the anode or the cathode electrode, with a higher current extracted by illuminating through the cathode. We locate a zone in the BHJ close to the anode with reduced photocurrent extraction due to a high electron trap density leading to

trap-assisted recombination in the active layer. We show that by reducing the time of ambient exposure we can reduce photocurrent asymmetry, i.e., degradation, and thereby attain all solution-processable PEDOT(PSS) solar cells with a performance that is comparable to sequentially cast devices.

RESULTS AND DISCUSSION

The PEDOT(PSS) electrodes are coated in ambient on polyethylene terephthalate (PET) substrates using a roll-to-roll slot-die coater. AL1 is spin-die or slot-die (bench top) coated on top of the anode stack PET/PEDOT(PSS) and AL2 on the cathode stack PET/PEDOT(PSS)/PEI. In Fig. 1a the roll lamination method using flexible substrates with the active layer as the adhesive is illustrated. Here, we demonstrate the method using two identically processed BHJ active layers composed of TQ1:PC₇₁BM laminated with a roll-to-roll laminator at 120 °C in ambient atmosphere.

Typical dark and 1 sun illumination current density vs. voltage (JV) curves from a laminated TQ1:PC₇₁BM based solar cell are shown in Fig. 1b together with an all-solution cast TQ1:PC₇₁BM device before and after a high reverse bias pulse to remove the shunting behavior.¹¹ The active layers of both devices were spin-coated in a nitrogen glove box. The performance of the laminated device shows identical V_{OC} but a slightly lower J_{SC} and FF compared to the sequentially solution-cast device following an electrical pulse. The dark current of the laminated device confirms that the problems with shunted devices in solution deposited PEDOT(PSS) based solar cells are completely avoided using active layer lamination. The laminated device is also not affected by subsequent reverse bias electrical pulses (not shown). Lamination thus allows the construction of non-shunted solar cells with PEDOT(PSS) as both the top and the bottom electrode. Since the solar cell JV curves are unaffected by electrical pulses, we conclude that shunting is not intrinsic to PEDOT(PSS), but is instead due to the deposition method.

Lamination sets the adhesive layer thicknesses to between 50 and 200 nm. Thus, this active layer lamination method is sensitive to dust particles, which are commonly in the micrometer size range. Nevertheless, laminated devices, prepared in our regular lab, i.e., not in a clean room, are functioning with satisfactory mechanical adhesion and good electrical contact. To address the details of lamination homogeneity photoluminescence (PL), optical microscopy and photocurrent imaging have been used, shown in the supporting information Supplementary Fig. S11, 2.

Unfortunately, ambient processing causes degradation, manifested as an asymmetry in the extracted photocurrent, with a higher photocurrent extracted by illuminating the solar cell through the cathode side than through the anode side. Below we investigate the origin of this degradation to show that it can be avoided by limiting the exposure time to ambient.

In the laminated semitransparent devices the extracted photocurrent differs significantly depending on the side of illumination. The JV curves show a ~45% higher photocurrent when the laminated cell processed in ambient conditions is illuminated from the cathode side (Fig. 2). Investigations of optical transmission, reflection and absorption have clarified that parasitic absorption in the electrodes is not the cause of this large asymmetry in the extracted photocurrent (Supplementary Fig. S13). The presence of a thin ~5–10 nm layer of PEI on the cathode only weakly affects the optical power dissipated in the device stack, and cannot account for the large variations observed. We have also confirmed these observations by calculating the optical power dissipation in the active layer with the transfer matrix method (TMM)²³ that shows a photocurrent difference of less than some few percent depending on what electrode is illuminated (Supplementary Fig. S14), irrespective of the active layer thickness. We also discard BHJ morphology variations induced by variations in surface energy of the PEDOT(PSS) and PEDOT(PSS)/PEI surfaces

from PL measurements. Both AL1 and AL2 are spectrally identical and also comparable in PL quenching efficiency, both for films prior to lamination and for the device excited through the anode and the cathode after lamination (Supplementary Fig. S15). The identical PL spectra and PL quenching indicate that charge generation should be comparable in AL1 and AL2. Thus, we conclude that neither optical in-coupling nor morphology-induced charge carrier generation differences are the main cause for the asymmetric photocurrent extraction.

To eliminate the influence of ambient conditions, it would be ideal to fully prepare the laminated devices in a controlled atmosphere, e.g., a glovebox filled with nitrogen. However, our current lamination tools are not suitable to do this. Instead, we have deposited the active layers under inert conditions and then immediately laminated them in ambient. We observe that the photoelectrical asymmetry is suppressed when the exposure to ambient is reduced. Figure 3 shows JV-curves from laminated TQ1:PC₇₁BM solar cells with the ALs spin-coated inert and then exposed to 60, 15 and 2 min of ambient atmosphere prior to lamination. The evolution of the asymmetry is clearly visible. Long-time exposure results in a reduction in the anode photocurrent over the full measured voltage range, whereas short exposure to ambient for 15 min leads to a pronounced s-kink in the JV curve. Short ambient exposure time (~2 min) eliminates the photocurrent asymmetry almost completely. Ambient exposure does not only degrade the photocurrent extraction upon illumination

through the anode, but also for illumination through the cathode. Sixty minutes of ambient exposure reduces J_{SC} by over 20% (Fig. 3c). Therefore, exposure to ambient conditions during production should be prevented to avoid degradation when producing organic solar cells by lamination using the AL as the adhesive.

Above we have shown that ambient processing and inert processing of the active layer with subsequent exposure to ambient conditions degrades the device. The loss of photocurrent is largest upon illumination through the anode. To avoid losses the active layer must be cast under inert conditions. However, inert processing of the active layer is not fully compatible with scalable and cost-efficient production methods needed for large area OPV. Hence, a deeper understanding of the cause of asymmetric photocurrent generation is required. Since we have already ruled out optical effects and problems with charge generation, the cause must be related to the transport and/or extraction of charge carriers.

Measurements of the external quantum efficiency (EQE) as function of wavelength show significant differences when the laminated cells are illuminated from different sides (Fig. 4a). As we have shown above, the difference in photocurrent generation and hence EQE cannot be understood from a purely optical point of view. A comparison of the absorption spectrum of the pure AL to the EQE spectrum indicates that carriers generated close to the anode do not contribute to the extracted photocurrent to the same degree as those generated close to the cathode (Fig. 4a). Upon illumination through the cathode, the EQE spectrum follows the absorption spectrum of the active material, while this is not the case for illumination through the anode (Fig. 4a). To understand these effects we have performed optical TMM simulations where a fraction of the active layer is assumed to have a considerably lower internal quantum efficiency (IQE).

Figure 4b shows a TMM simulated EQE spectrum of an active layer which has a dead zone from where no photocurrent is extracted (0% IQE), corresponding to 1/3 of the total AL thickness, stretching from the anode interface and into the device (Fig. 4c). The simulation is in good agreement with experiment (Fig. 4b). To probe the sensitivity of our simulations to different scenarios we have also extended the 0% IQE layer from the anode all the way to the lamination interface, corresponding to half of the laminated cell, i.e., the entire thickness of AL1. Assuming 75% IQE in AL2 results in an EQE spectrum that also gives an adequate fit to the experimental data. Thus, we conclude that charge carriers generated in an area stretching from the anode and possibly through the full thickness of AL1 cannot at all or can only partially be extracted.

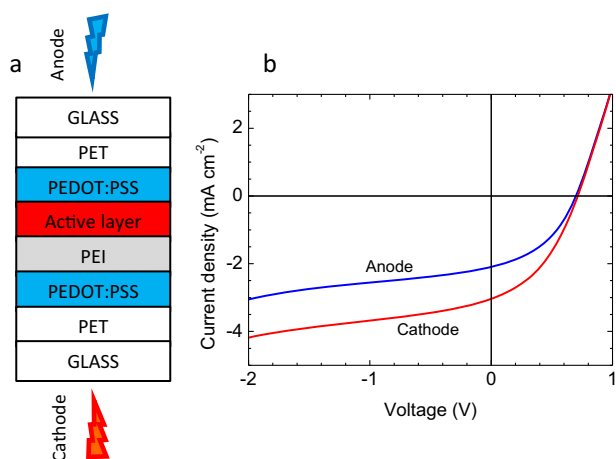


Fig. 2 Assymmetric photo current **a** Schematic of the laminated solar cell stack illuminated either through the anode (blue) or the cathode (red) side. **b** JV curves for an ambient-processed solar cell illuminated through the anode (blue) and through the cathode (red)

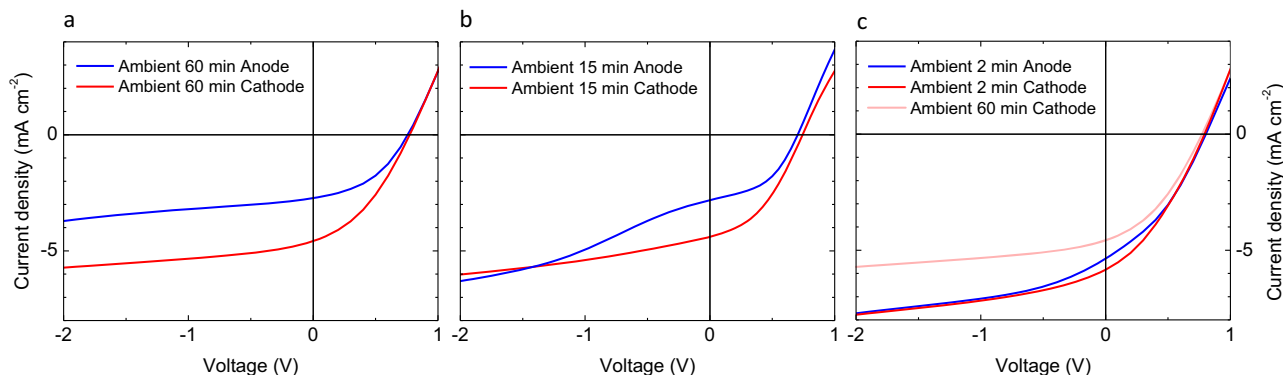


Fig. 3 Ambient exposure JV curves of laminated TQ1:PC₇₁BM solar cells (illuminated under one sun conditions) deposited under inert conditions and exposed to ambient for **a** 60, **b** 15 and **c** 2 min

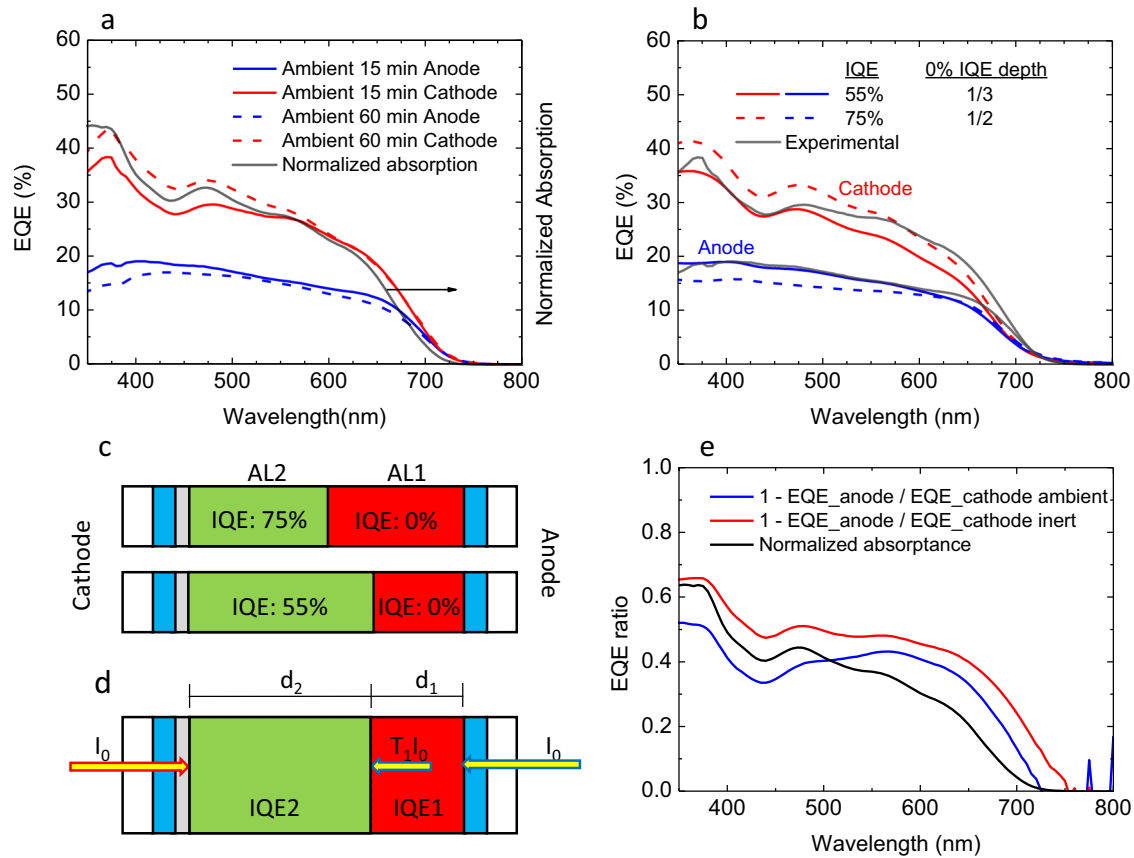


Fig. 4 Non contributing zone **a** Experimental EQE for devices exposed to ambient for 60 (dash) and 15 (solid) minutes illuminated through the anode (blue) and cathode (red), and the active layer absorption (black). **b** TMM simulated EQE for devices with 0% IQE on the anode half and 55% IQE on the cathode half (solid), 0% IQE in 1/3 of the device close to the anode and 75% IQE in the cathode side (dash) and experimental EQE from panel **a** (gray). **c** Illustration of the position of the areas with different IQE used in **b**. The active layer is divided in working (green) and dead (red) areas. **d** Schematic illustration of the $(1 - \text{EQE})$ ratio calculation. A part of the active layer with thickness d_1 and 0% IQE on the anode side reduces the illumination intensity incident upon the working part by transmission through the dead layer, T_1 . **e** $1 - \text{EQE}$ ratio for a device cast under inert conditions (red), a device exposed to ambient (blue) and the scaled absorption coefficient of the AL (black)

We further verify the presence of an area with $\text{IQE} = 0\%$ in AL1 by plotting the quantity $(1 - \text{EQE}_{\text{Anode}}/\text{EQE}_{\text{Cathode}})$ together with the normalized absorbance of the active layer (Fig. 4e). This ratio is comparable to the absorbance of AL1, as can also be shown analytically assuming $\text{IQE} = 0\%$ in AL1,

$$1 - \frac{\text{EQE}_{\text{Anode}}}{\text{EQE}_{\text{Cathode}}} = 1 - \frac{A_{A,2}\text{IQE}_2 + A_{A,1}\text{IQE}_1}{A_{C,1}\text{IQE}_1 + A_{C,2}\text{IQE}_2}$$

$$1 - \frac{A_{A,2}\text{IQE}_2}{A_{C,2}\text{IQE}_2} = 1 - \frac{A_{A,2}}{A_{C,2}} = 1 - \frac{A_2 T_1 l_0}{A_2 l_0} = 1 - T_1 = A_1 = \exp(-ad_1)$$

with $A_{X,i} = A_i/l_X$ where $A_{X,i}$ is the absorption in layer i by illuminating through side X , A_i the absorbance of layer i and l_X the light intensity incident on layer i . The two areas and the light paths are shown in Fig. 4d. Here, we assume equal light intensity incident from both sides and internal quantum efficiency equal to IQE_2 in the contributing part of the device, irrespective of illumination side. The absorption of the active layer resembling $(1 - \text{EQE}_{\text{Anode}}/\text{EQE}_{\text{Cathode}})$ implies that when illuminating from the anode side the generated charge carriers close to the anode are not collected, thus this part of the device acts as an optical filter where light is absorbed but does not allow for the extraction of photocurrent. Hence, from EQE measurements and optical simulations we conclude that the non-contributing zone is localized in AL1, stretching from the anode towards the lamination interface.

To investigate what causes the asymmetry, we use electrical measurements. The lower charge collection efficiency upon illumination from the anode suggests that for ambient-exposed devices the electron transport may be significantly less efficient than the hole transport. This is due to the fact that upon illumination from the anode side, more electrons (and holes) are photo-generated at the anode side, i.e., far away from the cathode, making it more difficult to reach the cathode electrode without recombining. We have studied electron-only and hole-only devices with the active layer immediately laminated and also exposed to ambient. This can be done in a very similar structure as the transparent PEDOT(PSS) electrodes can be used as selective electrodes for hole-only devices, and PEDOT(PSS)/PEI for electron-only devices. Space-charge-limited current (SCLC) measurements indicate that exposure to ambient has a dramatic effect on electron transport (Fig. 5a). The hole mobility of a laminated TQ1:PC₇₁BM 1:3 BHI is unaffected by ambient exposure, while electron transport deteriorates significantly. The devices behave identical both under forward and reverse bias, shown in Supplementary Fig. S16. The JV curve of the electron-only device behaves drastically different after ambient exposure. The characteristic shape of the degraded electron only JV curve ($J \propto V^6$) is as expected from a device with a very high trap density, as previously described by e.g., Blom et al.²⁴ Figure 5a thus implies the presence of a significant electron trap density after exposure to ambient.

The presence of electron traps may cause a loss of charge carriers to trap-assisted recombination. Blom et al. have shown

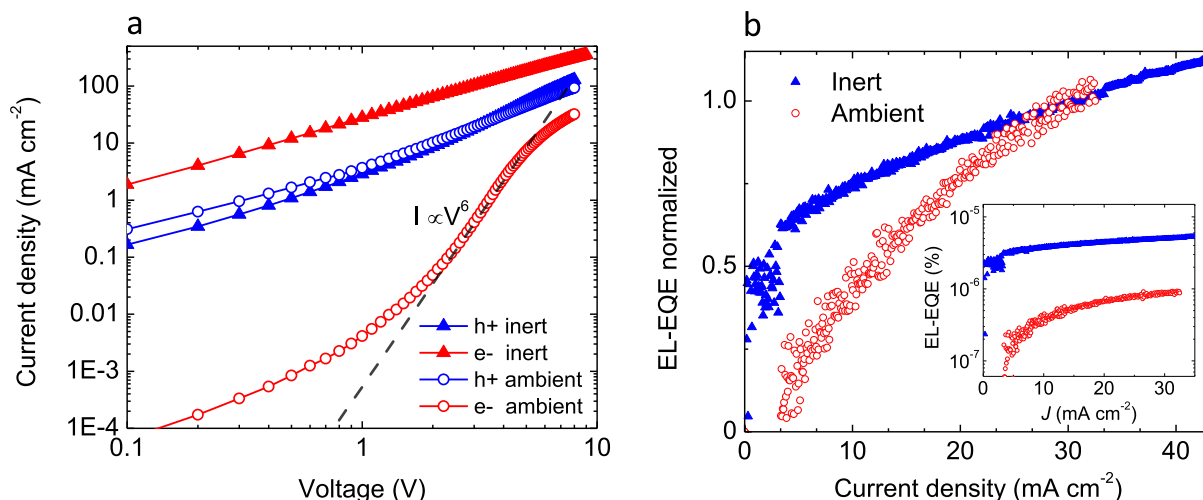


Fig. 5 Carrier transport **a** SCLC plot of laminated hole-only (blue) and electron-only (red) devices processed inert and exposed to ambient for 2 (filled triangles) and 45 (open circles) minutes. **b** EL-EQE vs. injected current-density for a laminated TQ1:PC₇₁BM device processed inert (blue triangles) and exposed to ambient for 45 min (red circles)

that voltage-dependent charge transfer (CT) electroluminescence quantum efficiency (EL-EQE) measurements can be used to identify the presence of non-radiative trap-assisted recombination.²⁴ The CT EL-EQE is bias-independent in the case of bimolecular recombination and is bias-dependent for trap-assisted recombination.²⁴ Figure 5b shows the measured EL-EQE for TQ1:PC₇₁BM devices processed under inert conditions and in ambient. We find that the ambient-processed device has an order of magnitude lower CT EL-EQE and a stronger dependence of CT EL-EQE on injected current density (Fig. 5b). Both observations are indicative of non-radiative trap-assisted recombination in the ambient-processed device. On basis of SCLC and EQE-EL measurements we conclude that exposure to ambient during lamination drastically deteriorates electron transport due to a strong increase of the electron trap density, giving rise to an increased trap-assisted recombination.

The significantly deteriorated electron transport at the anode may cause a pile up of electrons at the anode, i.e., space-charge. EL-EQE shows that trapped carriers recombine, but we also probe the presence of space-charge effects for photo-generated carriers by light-intensity-dependent photocurrent measurements (Fig. 6a). From $J_{SC} \propto I^a$, we fit a , where $a=1$ is an indication of weak bimolecular recombination, whereas $a=0.76$ originates due to space-charge effects.²⁵ We see no major differences in the slope as $a \sim 1$ for all devices, which means that space-charge effects do not play a significant role. This is in contrast to previous studies reporting asymmetric photocurrent extraction, which was attributed to space-charge effects.^{26–28} Thus, on basis of light-intensity dependent J_{SC} measurements we rule out space-charge effects as the origin of asymmetric photocurrent collection.

Due to the presence of electron traps in ambient exposed devices, the recombination mechanisms are expected to be different in AL1 and AL2, i.e., closer to the anode and the cathode electrodes, respectively. To investigate the difference in the recombination mechanisms at the two electrodes we have carried out light-intensity-dependent V_{OC} measurements (Fig. 6b). The slope (S) of the V_{OC} vs. illumination intensity can be used to determine the dominant recombination mechanism: $S=1$ for bimolecular recombination and $S > 1$ is typically assigned to trap-assisted recombination.²⁹ The increased slope for ambient-exposed devices is an indication of an increasing importance of trap-assisted recombination and we find a higher slope upon illumination from the anode, compared to the cathode. We have further confirmed that recombination is different depending on

what side is illuminated by a recently developed technique—intermodulation photocurrent (IMP),³⁰ see the Supporting Information Supplementary Fig. S17. From the light-intensity-dependent V_{OC} and IMP measurements we conclude that the amount of trap-assisted recombination increases upon ambient exposure and that it is more pronounced when illuminating from the anode side, as expected.

Our investigations have thus shown that the main cause of the asymmetric photocurrent collection are electron traps in the active layer, hampering electron transport through the device and causing trap-assisted recombination. However, we cannot explain the chemical nature of these traps. We speculate that the ambient-induced asymmetry could be caused by water molecules trapped in voids in the AL. Water molecules have been recently reported to induce traps in conjugated polymers for field-effect transistors,³¹ while oxidation of PCBM is known to also induce traps.³² A third possibility is that ambient exposure leads to a degradation of the lamination interface, where a region of electron traps is formed, blocking the extraction of electrons generated in AL1. We continue to investigate the degradation mechanisms leading to the increased electron trap density and we are currently searching for spectroscopic signatures of the presence of electron traps in the active layer.

To confirm the generality of our roll lamination method using the active layers as the adhesive we have also used a number of different donor:acceptor combinations. In all cases we have obtained good mechanical and electrical contact, albeit none of the BHJ systems have been optimized with respect to the PCE (Supplementary Fig. S18 and Supplementary Table S11).

The lamination method using AL as the adhesive opens up additional possibilities to tune the donor/acceptor stoichiometry of AL1 and AL2. We have found that using a pure PCBM layer as AL2 and a BHJ as AL1, the FF can be enhanced (FF ~ 0.40 vs. FF ~ 0.55). We have attained PCE = 2.6% with TQ1:PC₇₁BM and PCE = 3.5% with the commercial printing ink PI4 in semitransparent devices using this approach. The corresponding JV-curves are shown in Fig. 7. We have also added a paper reflector to the non-illuminated side of the semitransparent laminated solar cell¹¹ and reach PCE = 3.4% for TQ1:PC₇₁BM and PCE = 4% for PI4. This is the expected performance of PI4 in semitransparent PEDOT(PSS) devices and serves as a confirmation that our lamination method can yield as high performance as the sequentially cast devices.

We think that our lamination method could simplify the processing of ternary and quaternary blends as it may give a

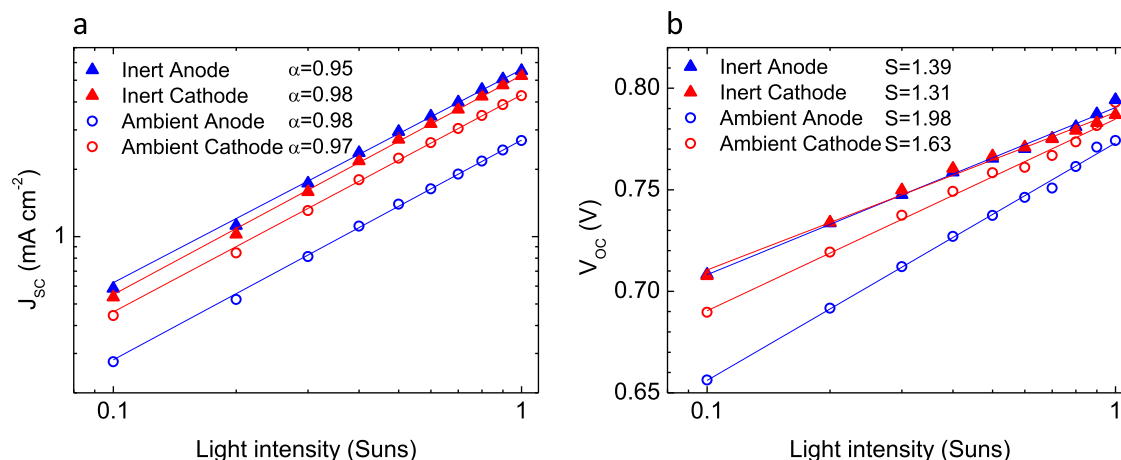


Fig. 6 Light intensity dependence **a** Short-circuit current and **b** open-circuit voltage as a function of light intensity for ambient (open circles) and inert devices (filled triangles), for devices illuminated through the cathode (red) and through the anode (blue) side. Solid lines are fits to $V_{oc} = S \cdot kT/q + m$ and $J_{sc} = I^\alpha$

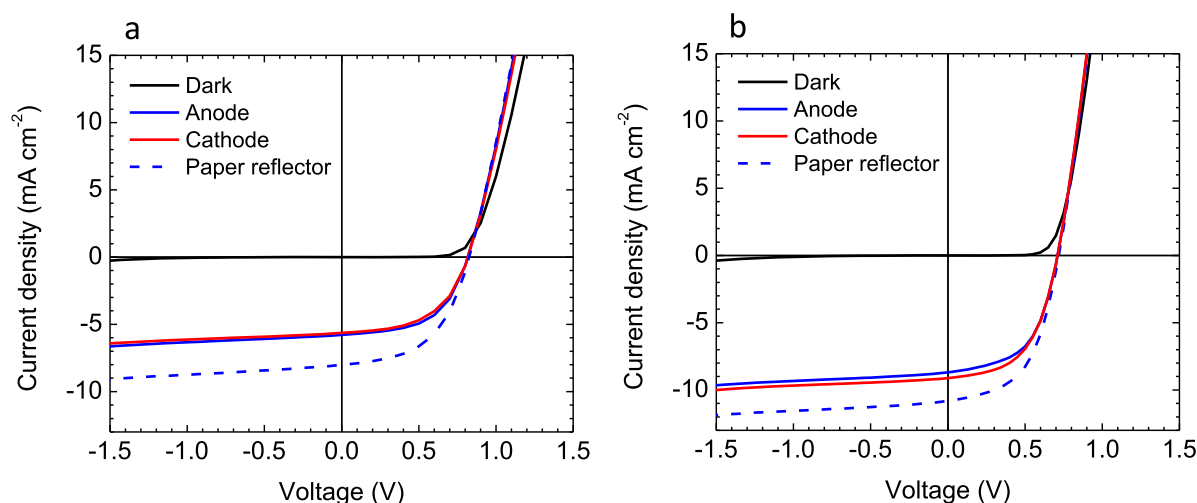


Fig. 7 BHJ/PCBM lamination JV-curves of **a** TQ1:PC₇₁BM and **b** PI4 laminated A1/A2 BHJ/PCBM semitransparent solar cells illuminated through the anode (solid blue), cathode (solid red) and with the addition of a white paper reflector (dashed blue, illumination from the anode side)

route to control and tune the vertical phase separation. We speculate that in a PCBM/BHJ laminate the fullerenes migrate across the lamination interface during the hot lamination (and possibly the polymer also) and this leads to a preferential gradient with PCBM enrichments closer to the cathode. Furthermore, using the lamination method the stoichiometry at either electrode can be tuned to optimize for favorable vertical phase separation. Thus, active layer lamination could open up a wide range of possibilities for organic solar cells, if the presence of electron traps is avoided, e.g., by processing in an inert environment.

CONCLUSIONS

In this work we have presented active layer lamination to construct all-organic solar cells, using the active layer as the adhesive layer. This results in well performing solar cells and solves the shunting problem common to devices fabricated using wet deposition of the top PEDOT(PSS) electrode. However, processing in ambient induces a strong asymmetry in photocurrent extraction. The asymmetry originates from ambient-induced electron traps, limiting electron extraction and increasing trap-assisted recombination. We have shown that degradation can

be avoided by limiting exposure to ambient. The use of solution-processed PEDOT(PSS) electrodes could thus enable truly low-cost OPV, where low material usage significantly reduces the environmental impact and helps improve the return of invested energy. We have shown that the lamination method is universal as we have laminated a number of different polymer:PCBM systems. The performance of the best of our laminated devices is comparable to those expected from sequential wet coating. We expect active layer lamination to be an important tool in future OPV production and research in the context of, i.e., improved control of vertical phase separation with tunable enrichment of the donor and/or the acceptor materials at either electrode.

METHODS

Laminated solar cells were constructed by laminating two stacks composed of PET/PEDOT(PSS)/PEI/Active layer to Active layer/PEDOT(PSS)/PET using a roll laminator (GSS DH-650S Graphical Solutions Scandinavia AB) with a roll temperature of 120 °C. The force on a force sensor (FlexiForce A201, Tekscan), pushed between the lamination rollers, was measured to ~30 N. The PEDOT(PSS) PH1000 (Heraeus) with 6 vol% ethylene glycol (Sigma) and 0.5 vol% Capstone FS-30 (Dupont) and 5–10 nm thick PEI cathode modification layer coated from isopropanol (1 g l⁻¹)

was slot-die coated (Solar X3, FOM Technologies) on 125 μm heat-stabilized PET Melinex ST505 (Tekra). More details on the R2R coating line are found in ref. ¹²

Active layers were either spin-coated or slot-die coated using a Mini Roll Coater (FOM Technologies) and coated either in a glove box (<10 ppm O_2 , H_2O) or in ambient. TQ1 was synthesized according to ref. ⁷ PC₇₁BM (99%) was bought from Solenne BV and PI4 ink was bought from infinityPV. For the spin coated devices, TQ1:PC₇₁BM active layers were spin coated at 500 rpm from 25 g l^{-1} orthodichlorobenzene solutions and were in total (after lamination) 150–200 nm thick. The PCBM layer on the cathode for the devices with PCBM/BHJ lamination interfaces were spin coated at 2000 rpm from a 10 g l^{-1} chloroform solution. PI4 active layers were spin coated at 450 rpm to an expected thickness of ~ 300 nm. The PI4 based devices were annealed after lamination and encapsulation at 110 $^\circ\text{C}$ for 5 min on a hot plate. The reported performance of PCBM/BHJ devices are here devices. The width of the laminated solar cells were typically 2 mm and they were carefully patterned on the PEDOT(PSS) anode with a scalpel in to ~ 5 mm wide cells prior to lamination. Photographs of laminated devices are shown in Supplementary Fig. S18. All samples were encapsulated with glass lids and UV-curing adhesive (Norland NOA73) after lamination. Silver paint (Agar AGG302) was added to the PEDOT(PSS) to reduce the contact resistance to the contacts.

Sequentially cast solar cells were constructed on pre-clean glass substrate. PEDOT electrodes were deposited at a spin-speed of 1500 rpm. The solution was prepared in the same way as that for the laminated devices. The active layer was spin-coated at 1000 rpm from solution based on TQ1:PCBM (1:2.5, weight ratio, oDCB). Solution concentration was 30 mg ml^{-1} . PEIE (0.05 wt% in IPA) was deposited at 2000 rpm. More details on the sequentially cast devices are found in ref. ¹¹

PL was measured using a Shamrock SR 303i spectrograph coupled to a Newton EMCCD silicon detector. The samples were excited using a 532 nm laser diode (CW 532-005, Roithner Lasertechnik).

J - V curves, V_{OC} and J_{SC} vs. light intensity were recorded with a Keithley 2400 Source Meter under AM 1.5 G illumination with a solar simulator (LSH-7320 LED Solar Simulator, Newport). Illumination intensity was controlled using neutral density filters and the preset intensity setting from 0.1 to 1.0 suns. V_{OC} and J_{SC} vs. light intensity measurements were confirmed on two different samples, single carrier device degradation after ambient exposure on at least four different samples and solar cell degradation upon ambient exposure on at least twenty samples. To confirm that no cross talk occurred between cells the resistance between cells was controlled with a multimeter and the neighboring cells were also masked.

EQE was measured using a QE-R (Enlitech). Photocurrent imaging (LBIC) was measured by scanning a red laser diode with 635 nm emission wavelength (LDM635/5LTM from Roithner Lasertechnik GmbH) over the solar cell surface. The spot size was ~ 100 μm and the diode was modulated at 3.5 kHz using a multifrequency lock-in amplifier (IMP 2-32 from Intermodulation Products AB). The photocurrent response was passed through a pre-amplifier (SR570 from Stanford Research Systems) before being recorded by the lock-in amplifier. To scan the solar cell surface a XY micrometer precision translation table (Nanomotion II, Melles Girot) was controlled and synchronized with the software IMP Suite (Intermodulation Products AB).

UV-Vis was measured with a Lambda 950 UV-Vis (Perkin Elmer). TMM calculations were performed using an in house Matlab script with optical constants from the local library of optical constants, except for PEI. The optical constants of PEI spin-coated on a silicon substrate was determined using spectroscopic ellipsometry with an RC2 ellipsometer from J.A. Woollam Co., Inc. (USA) and the software CompleteEASE (J.A. Woollam Co., Inc.). Modeling was performed with B-splines.

IMP was measured with two 635 nm laser diodes (LDM635/5LTM from Roithner Lasertechnik GmbH) modulated at 3.5 (vertical incidence) and 6 kHz (20° incidence angle) using an intermodulation lock-in amplifier (IMP 2-32 from Intermodulation Products AB). The photocurrent response of the solar cell was pre-amplified (SR570 from Stanford Research Systems) before measured with the lock-in amplifier. The software IMP Suite (Intermodulation Products AB) was used to control the lock-in amplifier. To confirm overlap between the both illumination spots the 2nd order IMP signal was monitored live while one laser was carefully moved using a XY manual micrometer stage until the signal reached a maximum.

EQE EL was measured using a homebuilt system using a large-area calibrated Siphodiode from Oriel.

Data availability

All relevant data are available from authors upon reasonable request.

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AUTHOR CONTRIBUTIONS

J.B. and T.Ö. contributed equally to this work. J.B. planned the experiments, measured photocurrent imaging, side-dependent PL, IMP, made the TMM calculations, analyzed the data, and wrote the manuscript. T.Ö. made all laminated devices, measured JV, UV-Vis and SCLC, contributed to the analysis and revised the manuscript. A.M. measured EL-EQE, contributed to the analysis of the data and revised the manuscript. L.E.A. measured EQE and V_{OC} and J_{SC} vs. illumination intensity and revised the manuscript. Z.T. made the sequentially cast solar cells, contributed to the method development and revised the manuscript. W.C. contributed to the method development and revised the manuscript. Z.M. measured PL during lamination and revised the manuscript. D.G. and M.A. contributed with the upscaled synthesis of TQ1 and P3TI polymers used and revised the manuscript. M.K. contributed to the data analysis and discussions and revised the manuscript. O.I. initiated the project and contributed to analysis and discussions and writing and revising the manuscript.

ADDITIONAL INFORMATION

Supplementary information accompanies the paper on the *npj Flexible Electronics* website (<https://doi.org/10.1038/s41528-017-0017-6>).

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